



High-performance NO₂ gas sensor based on reduced graphene oxide/ZrO₂ hybrids

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Abstract

The increasing concern over environmental pollution, particularly from nitrogen dioxide (NO₂) emissions, necessitates the development of efficient NO₂ detection sensors. This study introduces reduced graphene oxide (rGO)/ZrO₂ hybrids for enhanced NO₂ gas sensing. Utilizing a modified Hummer's method, graphene oxide (GO) flakes were synthesized and subsequently sputtered with 10 nm ZrO₂ film, followed by thermal annealing to produce rGO/ZrO₂ hybrids. The hybrids were characterized using various techniques including SEM, TEM, AFM, Raman spectroscopy, and XRD, confirming successful synthesis and reduction of GO, as well as the formation of ZrO₂ nanoparticles. Gas sensing tests revealed superior sensitivity to NO₂ in the hybrids due to efficient electron transfer between rGO and ZrO₂, resulting in increased hole concentration in rGO and enhanced conductivity. The cyclic performance of the hybrids showed stable response and recovery to NO₂, while selectivity tests demonstrated high sensitivity to NO₂ over other gases including NH₃, ethanol, and oxygen. This study highlights the potential of rGO/ZrO₂ hybrids as high-performance NO₂ gas sensors, offering promising prospects for environmental monitoring and public health protection.

1. Introduction

Rapid population growth drives the need for economic expansion to support livelihoods. However, this growth often leads to environmental challenges, particularly concerning air pollution [1]. Industrial activities are a leading cause of air pollution, particularly in the emission of nitrogen dioxide (NO₂) [2]. This hazardous gas, emitted primarily from vehicular exhausts, power plants, and industrial facilities, poses grave risks to human health and ecosystems, triggering respiratory ailments and exacerbating environmental degradation [3]. Motivated by the urgent need to protect public health and address environmental harm, researchers are increasingly focused on developing better NO₂ detection technologies. Among different techniques, chemiresistors gas sensors have received considerable attention for their simplicity, cost-effectiveness, and scalability [4]. These sensors operate based on changes in electrical resistance upon exposure to target gases [4]. However, selecting the right material for the sensor's channel is crucial for optimal performance.

Even though graphene was discovered in 2004, it remains a highly promising material due to its exceptional properties [5]. This two-dimensional carbon allotrope possesses remarkable physical and chemical properties that make it suitable for a variety of applications, especially for NO₂ gas sensors [3]. Graphene-based chemiresistors offer numerous advantages for NO₂ detection, including reasonable sensitivity, fast response times, and low power consumption [6]. However, when compared to commercial NO₂ sensors, Graphene-based chemiresistors often exhibit lower sensitivity and performance levels. To address these challenges, researchers offer hybrid materials by combining graphene with other nanomaterials. While a variety of nanomaterials are being explored for this purpose, metal oxides stand out

as particularly promising candidates due to their high surface area, chemical reactivity, and tunable properties [7]. Up to now, different Graphene/metal oxide structures like Gr/ZnO [8], Gr/SnO₂ [9], Gr/WO₃ [10], Gr/In₂O₃ [11], Gr/NiO [12], and so on are introduced for NO₂ sensing. To the best of our knowledge, no reports have been made on the Zirconium oxide (ZrO₂) nanoparticles for the NO₂ detection. Although individual ZrO₂ nanoparticles are not suitable for this case since it has a very low electrical conductivity, Gr/ZrO₂ hybrids can exhibit promising potentials for enhancing NO₂ sensing capabilities.

In this study, we introduce reduced graphene oxide (rGO)/ZrO₂ hybrids as a novel approach for efficient NO₂ gas sensing. ZrO₂ presents several advantages that make it a promising candidate for NO₂ sensing, including high thermal stability, large surface area, excellent catalytic properties, and chemical inertness [13,14]. These properties can significantly enhance the sensor's performance by providing more active sites for gas adsorption and maintaining structural integrity at high temperatures. Our group is the first to introduce ZrO₂ in combination with reduced graphene oxide (rGO) for NO₂ detection. While our initial findings are promising, further investigations are needed to fully understand and optimize this novel material system. Through a comprehensive synthesis and characterization process involving modified Hummer's method for graphene oxide (GO) synthesis, sputtering of ZrO₂ film, and thermal annealing, we successfully demonstrate the formation of rGO/ZrO₂ hybrids with enhanced NO₂ sensing capabilities. Our investigation reveals efficient electron transfer between rGO and ZrO₂, leading to heightened sensitivity to NO₂, stable cyclic performance, and high selectivity over other gases. This research not only pioneers the development of high-performance NO₂ gas sensors but also offers promising prospects

for their application in environmental monitoring and public health protection.

2. Experimental

2.1 Material

Natural Graphite powder (100 mesh) was acquired from Fluke. Sodium nitrite (99%), sulfuric acid (95% to 98%), potassium permanganate (99%), and hydrochloric acid (37%), hydrogen peroxide (30%) were purchased from Sigma Aldrich. Zirconia - Sputter Target (3 N to 4 N purity) was supplied from Evochem.

2.2 Characterizations

Field emission scanning electron microscopy (Hitachi-S4160) was used to take the morphology of samples. Transmission electron microscopy (CM30m Philips) was utilized to investigate the successful synthesis of GO sheets. Raman spectra were measured by a confocal Raman microscope (Bruker Senterra, 532 nm excitation laser). X-ray diffraction patterns were obtained by PANalytical X'Pert Pro MPD diffractometer. Thickness profiles were taken by atomic force microscopy (AFM) via NOVA NT-MDT SOLVER NEXT, Russia. The NO₂ gas sensing tests were done by a home-made setup including a stainless-steel chamber with connections for electrical measuring, and mass flow controllers which were connected to the Keithley source measure unit (2400, SMU).

2.3 Synthesis of GO sheets

Natural graphite powder (1 g) and sodium nitrate (0.5 g) were mixed in sulfuric acid (20 mL) in an ice-water bath. Then, potassium permanganate (2 g) was gradually added to the mixture and stirred for 60 min. After that, DI water (40 mL) and hydrogen peroxide (200 mL) were added to the mixture, followed by further diluting with hydrochloric acid. The resulting dispersion was centrifuged at 4000 rpm for 2 h and sediments were collected in DI to experience another round of centrifugation. Finally, the remaining sediments were collected in DI and subjected to bath sonication for 2 h resulting in exfoliation of graphite oxide flakes. Then, the dispersion was centrifuged at 2000 rpm for 2 h and the top supernatant was collected as GO sheet dispersion. To prepare the GO film, a few drops of GO dispersion were cast on the quartz substrate and let dry overnight.

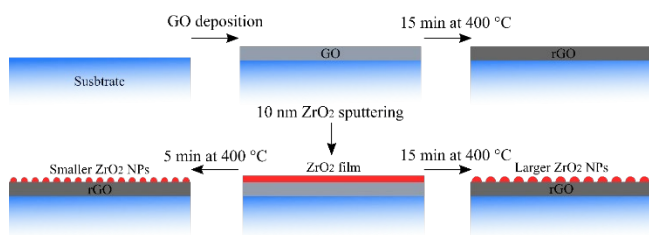


Figure 1. Sensor fabrication process for rGO, 5 min-treated rGO/ZrO₂, and 15 min-treated rGO/ZrO₂ sensors.

2.4 Preparation of rGO/ZrO₂ hybrid

GO/quartz samples were placed in the RF sputtering system and 10 nm ZrO₂ was deposited on them. Then, samples were placed in the quartz tube and annealed at 400°C for 5 min and 15 min under 100 sccm argon flow.

3. Result and discussion

3.1 SEM/TEM characterization of GO sheets

GO sheets are prepared using a modified Hummer's method according to the experimental section [15]. This method is easy and high throughput. Figure 2 (a) shows the morphology of the GO sheets on the substrate. The good concentration of GO flakes allows uniform deposition of them over the substrate. In the context of utilizing GO as the channel of NO₂ sensor, maintaining a good concentration of GO flakes is crucial. An optimal concentration (here 4 mg·mL⁻¹) ensures a uniform deposition of GO flakes on the substrate, which is vital for forming a consistent and homogeneous sensing channel. This uniformity is essential for the device's performance, as it enhances the sensitivity and accuracy of NO₂ detection by providing a stable and continuous electrical pathway. Inconsistent deposition could lead to variability in the device's response and compromise the reliability of NO₂ detection. Figure 2(b)) exhibits the higher magnified image of the GO film. Moreover, the TEM image of the GO sheets is shown in Figure 2(c). The layered structure of the GO sheets can be seen in the image. They also show good transparency which refers to efficient exfoliation of the bulk graphite powder.

3.2 Effect of thermal treatment on the formation of rGO/ZrO₂ hybrid

After depositing GO film on the quartz substrate, 10 nm ZrO₂ film was sputtered on them using an RF sputtering system. The thermal treatment is used for two reasons. First, this process promotes the reduction of GO film into the rGO film. Second, thermal treatment helps to convert ZrO₂ films into nanoparticles. In the latter, two annealing times of 5 min and 15 min were investigated to see their effect

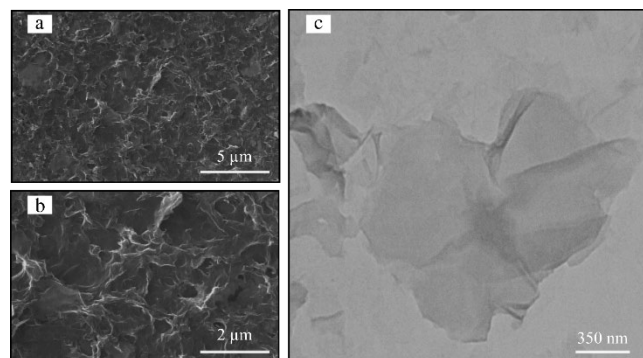


Figure 2. (a) SEM image of the deposited GO film, (b) higher magnification of its surface, and (c) TEM image of the GO sheets.

on the final formation of the hybrids. Figure 3(a-b) shows the SEM image of the hybrid that remained for 5 min at 400°C. The size distribution histogram in Figure 3(c) demonstrates the average size of 39 nm that is distributed in the 20 nm to 60 nm size variation. When samples remained longer (15 min) in the same condition, it was found that it led to a higher average size of 42 nm for the nanoparticle with a wider size distribution (Figure 3(d-f)). This may originate from a combination of more nanoparticles due to their thermal motion during thermal treatment [16].

3.3 Characterization of surface roughness in rGO/ZrO₂ hybrid films via AFM measurements

To see the roughness of the samples, AFM measurements were taken in the non-contact mode. Figure 4(a) exhibits the 3D uniform morphology of GO film with a maximum roughness of 300 nm. In the case of a 5 min thermal treated sample (Figure 4(b)), it has a higher roughness around 500 nm which may be due to the formation of nanoparticles. The AFM image of the 15 min treated sample (Figure 4(c)) demonstrates a higher roughness around 1000 nm that is related to the creation of large nanoparticles in this sample according to the SEM images.

3.4 Electrical conductivity characterization of rGO/ZrO₂ hybrid

To investigate the electrical conductivity of the samples, IV measurements were done by adding silver paste to the samples. Figure 5(a) reveals the IV testing of the samples which is done in the voltage range of -3 V to +3 V and the corresponding currents were measured via the Keithley source measure unit [17]. Accordingly, the GO film shows high resistance with a too-poor electrical current. The 5 min treated sample shows better electrical conductivity and the highest electrical conductivity was measured for the 15 min treated sample. The measured resistance of the samples is shown in Figure 5(b) in which values of 12.1 MΩ, 340 KΩ, and 210 KΩ are calculated for the GO, 5 min, and 15 min treated samples, respectively. The increase of electrical conductivity of two later samples is due to the thermal reduction of GO film into the rGO film [18]. As a result, the 15 min treated film shows higher conductivity since it remains longer time in the 400°C temperature and is subjected to more thermal reductions.

3.5 Raman and XRD characterizations of the thermally formed rGO/ZrO₂ Hybrid

The thermal reduction of the GO film can also be investigated by Raman spectroscopy. Figure 6(a) shows the Raman spectra of the films. Based on it, the GO film has two Raman peaks around 1351 cm⁻¹ and 1600 cm⁻¹ that are attributed to D and G phonon oscillations [19]. The same peaks are also observed for the 5 min and 15 min treated films. The ratio of I_D/I_G peak is used to show the degree of GO reduction into the rGO in Figure 6(b) [15]. Based on it, the values of 0.96, 0.86, and 0.83 are calculated for the GO, 5 min, and 15 min treated films, respectively. A reduction in the I_D/I_G ratio is due to the decrease of

oxygen-containing functionalized groups in the rGO in comparison with GO sheets [18]. To further investigate this reduction and prove the formation of the ZrO₂ nanoparticles, XRD spectra of the GO and 15 min treated film were provided in Figure 6(c). GO film shows one broad peak around 9.8° that is related to the [001] crystal orientation [20]. In the case of the 15 min treated samples, one strong peak is observed around 24.3° which corresponds to the [002] crystal orientation of the rGO structure and proves its reductions [20]. In addition to this peak, five other peaks appeared around 28.1°, 32.5°, 38.2°, 50.0°, and 58.6° that are related to the [-111], [111], [120], [002], and [131] crystal orientations of the ZrO₂ structure, respectively [21].

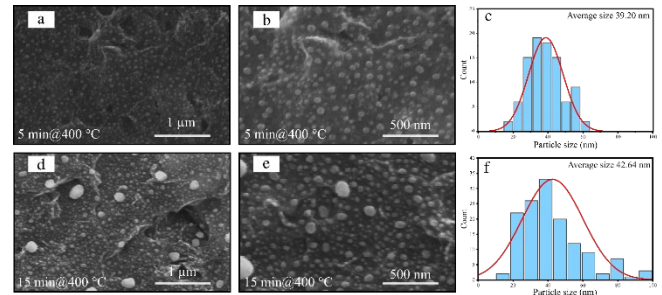


Figure 3. SEM image of hybrids after (a, b) 5 min, and (d-e) 15 min thermal annealing at 400°C. Size distribution histogram of the nanoparticles after (c) 5 min, and (f) 15 min thermal annealing at 400°C.

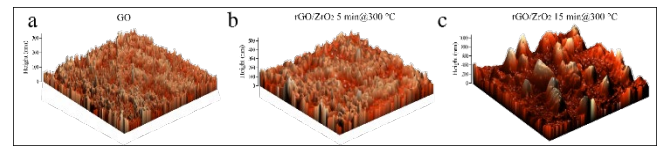


Figure 4. AFM 3d profile of (a) GO, (b) 5 min, and (c) 15 min treated samples at 400°C.

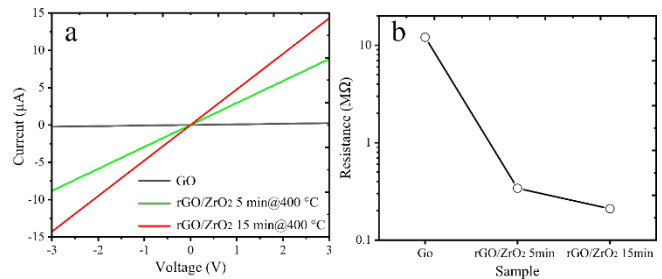


Figure 5. (a) I-V test, and (b) the corresponding resistance of the GO, 5 min, and 15 min treated films.

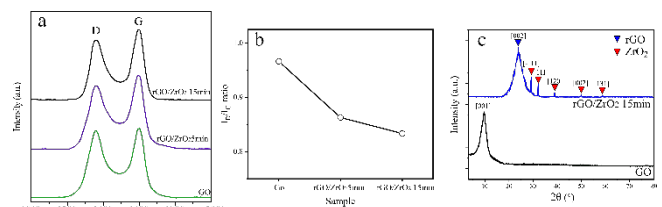


Figure 6. (a) Raman spectra and, (b) corresponding calculated I_D/I_G ratio of the GO, 5 min, and 15 min treated films, and (c) XRD patterns of the GO, and 15 min treated films.

3.6 NO₂ sensing performance of rGO/ZrO₂ hybrid sensors and selectivity analysis

The prepared samples then were used to test NO₂ detection in a home-made setup connected to the SMU system as shown in Figure 7.

The sensitivity is calculated according to the following Equation [3]:

$$S(\%) = \frac{R_{\text{gas}} - R_{\text{air}}}{R_{\text{air}}} \times 100\% \quad (1)$$

Where R_{gas} is the resistance of the sensor upon exposure to NO₂ and R_{air} the resistance of the sensor upon exposure to air. rGO is considered a p-type semiconductor which means its electrical conductivity is dominated by holes [4]. When NO₂ molecules are absorbed on its surface, they extract electrons from the rGO and increase the density of holes in the rGO film [22]. As a result, the overall resistance of the rGO film is decreased upon exposure to the NO₂ gas. The same behavior is observed for the 5 min and 15 min treated rGO/ZrO₂ samples since the rGO is dominant in these samples. According to Figure 8(a), the sensitivity of -24%, -35%, and -28% is measured for the rGO, 5 min, and 15 min treated rGO/ZrO₂ samples under 100 ppm NO₂ concentration, respectively. The rGO/ZrO₂ samples show higher sensitivity compared with the pure rGO due to the electron transfer that happened between them. This electron transfer occurred because the rGO (4.9 eV) [23] has a higher work function than ZrO₂ (3.1 eV) [24], forming a Schottky-type junction between them. Exposure to NO₂ caused electron transfer from ZrO₂ to NO₂, decreasing ZrO₂ carrier concentration [25]. As a result, electrons are transferred back from rGO to ZrO₂, resulting in higher hole concentration in rGO. This led to increased conductivity in rGO, resulting in enhanced sensitivity to NO₂ compared to pure rGO film [25]. Figure 8(b) demonstrates the sensitivity of the samples as a function of different NO₂ concentrations. In rGO, the sensitivity is increased from 6.7% to 24% upon exposure to the NO₂ concentrations from 5 ppm to 100 ppm, respectively. For 5 min treated rGO/ZrO₂, sensitivity shows the value of 15.8% in 5 ppm NO₂ and 35% in 100 ppm NO₂ concentration which are the highest values among two other samples. In the case of 15 min treated rGO/ZrO₂, sensitivity is increased from 9.5% to 28% with increasing the NO₂ concentration from 5 ppm to 100 ppm, respectively. The higher sensitivity of the 5 min treated sample is due to its smaller size nanoparticles that provide a larger surface area and more active sites for absorbing NO₂ molecules [26,27].

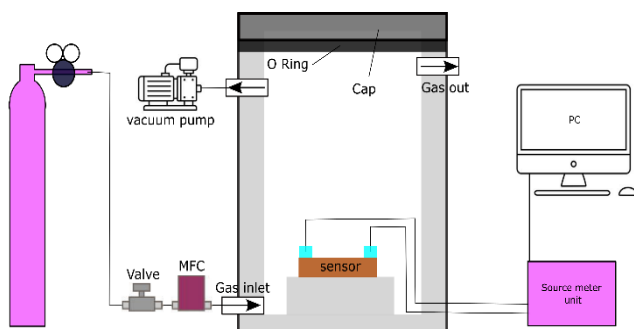


Figure 7. Schematic of the homemade setup for testing the rGO/ZrO₂ sensor.

The cyclic performance of the 5 min treated rGO/ZrO₂ sample is illustrated in Figure 8(c) upon exposure to 100 ppm NO₂ concentration. Based on it, the sample shows a very stable performance both in responding to the gas and recovering to its baseline. Moreover, the selectivity of the sensor was investigated for three other gas species including NH₃, ethanol (C₂H₅OH), and oxygen (O₂). The sensor's responses to NH₃, ethanol, and oxygen alongside NO₂ stem from the rGO's surface interactions with various gas molecules. NO₂, due to its strong oxidizing nature, exhibits the highest sensitivity, whereas NH₃ and ethanol adsorb with lower affinity, causing minor resistance changes. Oxygen interacts by withdrawing electrons, altering rGO conductivity [3,6,11,12,14,22]. According to Figure 8(d), the sensitivity values of -35%, 21%, -4%, and 9% are measured for the 100 ppm concentration of NO₂, NH₃, ethanol, and O₂, respectively. based on it, the highest sensitivity is measured for the NO₂ gas which shows the good performance of the sensor in detecting NO₂ gas. The proposed sensor responds to O₂ as well because oxygen molecules can interact with the rGO surface, altering its electrical properties and thus affecting the sensor's response. Oxygen is a common interfering gas that can influence the conductivity of rGO due to its electron-withdrawing nature, leading to changes in the resistance of the sensor [28]. To address the interference of oxygen in the selectivity of the NO₂ detection, several strategies can be employed. Firstly, surface functionalization of the GO layer with specific groups that have a higher affinity for NO₂ can enhance selectivity [29]. Additionally, incorporating selective coatings or membranes that permit the passage of NO₂ while blocking O₂ molecules can effectively reduce interference [30]. Optimizing the operational temperature and introducing chemical dopants to the GO structure are also viable options to improve selectivity [28].

3.7 Comparative analysis and advantages of rGO/ZrO₂ hybrid sensor for NO₂ detection

The summary of the sensor performance is provided in Table 1 and compared with some sensors reported in other studies. All of them operate at room temperature. Accordingly, the suggested sensor shows very good performance compared with other sensors. In comparison to the WO₃ nanoparticle-loaded MWCNT-rGO hybrids that offer higher sensitivity, the proposed rGO/ZrO₂ hybrid demonstrates distinct advantages, particularly in terms of simplicity, time efficiency, and cost-effectiveness. The rGO/WO₃/MWCNT sensor necessitates a multi-step synthesis process involving the preparation of WO₃ nanoparticles, MWCNT-rGO hybrids, and subsequent sensor fabrication, while our method streamlines the process by directly synthesizing rGO/ZrO₂ hybrids through a straightforward procedure. The synthesis involves the sputtering of ZrO₂ film onto GO/quartz substrates followed by thermal annealing, eliminating the need for complex precursor materials and lengthy synthesis steps. Moreover, the materials used in our synthesis are readily available and cost-effective, further contributing to the overall affordability and accessibility of the sensing platform. Thus, our method presents a promising alternative for efficient NO₂ gas sensing, offering the advantages of simplicity, reduced processing time, and cost-effectiveness, which are essential considerations for scalable sensor fabrication and widespread deployment in real-world applications.

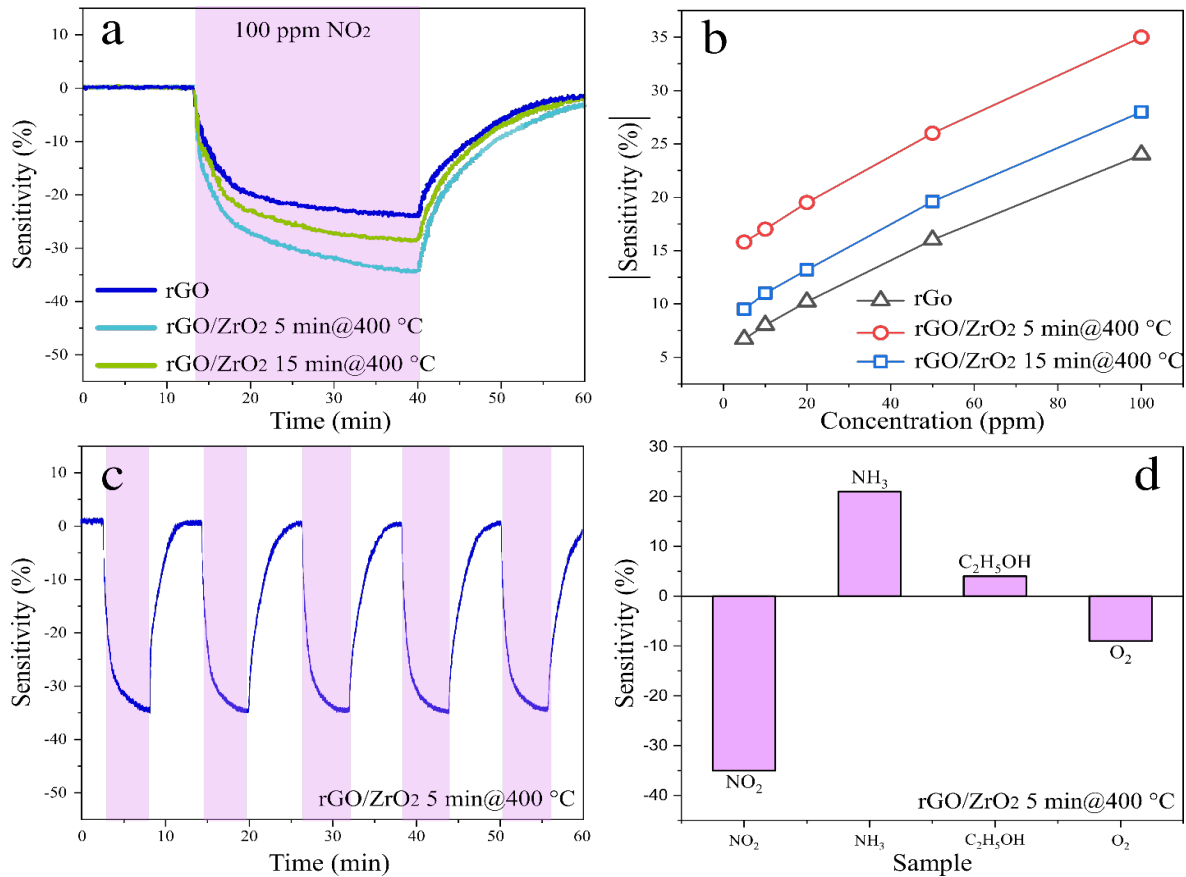


Figure 8. (a) sensitivity of the samples upon exposure to 100 ppm NO₂ gas, (b) sensitivity of the sensors as a function of NO₂ concentrations, (c) cyclic performance of the 5 min treated rGO/ZrO₂ sensors under 100 ppm NO₂ concentration, and (d) selectivity of the 5 min treated rGO/ZrO₂ to NO₂, NH₃, ethanol, and O₂ gas species.

Table 1. The comparison of NO₂ detection performance of rGO/ZrO₂ hybrid with other studies at room temperature testing.

Sensor's material	NO ₂ concentration	Sensitivity (%)	Ref.
rGO/ZrO ₂	5 ppm	15.8	Present
rGO/ZnO	5 ppm	14.2	[8]
rGO/SnO ₂	5 ppm	3.7	[9]
rGO/WO ₃ /MWCNT	5 ppm	17	[10]
rGO/In ₂ O ₃	30 ppm	8.25	[11]
rGO/NiO/BiVO ₄	2 ppm	4.3	[12]

4. Conclusions

In conclusion, the study successfully demonstrated the synthesis and characterization of reduced graphene oxide (rGO)/ZrO₂ hybrids for NO₂ gas sensing applications. The hybrids exhibited enhanced sensitivity to NO₂ gas compared to pure rGO, owing to the electron transfer mechanism between rGO and ZrO₂. The thermal annealing process not only facilitated the reduction of GO to rGO but also led to the formation of ZrO₂ nanoparticles, contributing to improved gas sensing properties. The superior sensitivity, stable cyclic performance, and high selectivity of the rGO/ZrO₂ hybrids towards NO₂ highlight their potential for practical applications in environmental monitoring and public health protection. Further optimization of synthesis parameters and exploration of other hybrid materials could lead to even higher-performance gas sensors for various environmental pollutants. This research contributes to the advancement of NO₂ detection technologies and underscores the significance of hybrid materials in enhancing gas sensing capabilities.

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